



# Ultrafine Iridium Oxide Nanorods Synthesized by Molten Salt Method toward Electrocatalytic Oxygen and Hydrogen Evolution Reactions



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## ABSTRACT

Ultrafine iridium oxide nanorods (IrO<sub>2</sub> NRs) were successfully synthesized using a molten salt method at 650 °C. The structural and morphological characterizations of these IrO<sub>2</sub> NRs were carried out by powder X-ray diffraction, Raman spectroscopy and electron microscopic techniques. Compared to commercial IrO<sub>2</sub> nanoparticles (IrO<sub>2</sub> NPs) and previous reports, these IrO<sub>2</sub> NRs show enhanced electrocatalytic activity to oxygen and hydrogen evolution reactions by passing either N<sub>2</sub> or O<sub>2</sub> gas in a 0.5 M KOH electrolyte before electrochemical measurements, including cyclic voltammetry, chronoamperometry and electrochemical impedance spectroscopy. Specifically, the current densities from the as-synthesized IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs were measured in 0.5 M KOH electrolyte to be 70 and 58 (OER, deaerated, at 0.6 V versus Ag/AgCl), 71 and 61 (OER, O<sub>2</sub>, from −0.10 to 1.0 V versus Ag/AgCl at 50 mV/s), and 25 and 14 (HER, deaerated, at −1.4 V versus Ag/AgCl) mA/cm<sup>2</sup>, respectively. These results are comparable with, and in most cases, higher than reported data in the literature. Therefore, the current study reports not only a novel synthetic process for IrO<sub>2</sub> but also a high efficient IrO<sub>2</sub> nanostructure, and it is expected that these IrO<sub>2</sub> NRs can serve as a benchmark in the development of active OER and HER (photo)electrocatalysts for various applications.

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## 1. Introduction

The storage of renewable energy using hydrogen “fuel” has been viewed as a viable mechanism by electrolysis of water into oxygen and hydrogen fuels, but is hampered by the slow kinetics of the oxygen evolution reaction (OER). Hence there is a broad effort to improve performance of currently used materials and develop new materials. Iridium oxide (IrO<sub>2</sub>) is an effective stable electro-catalyst which can lower the over-potential and keep a remarkable current to improve the efficiency of fuel generation technologies [1,2]. Nanostructured IrO<sub>2</sub> materials have been highlighted as the most promising electrode materials to enhance the electrocatalytic activity in both OER and hydrogen evolution reaction (HER) from the photo-electrochemical water-splitting and electrolysis [3,4,5,6,7,8]. Nanocrystalline IrO<sub>2</sub> and RuO<sub>2</sub> are recognized to be the best electrocatalysts for HER and OER in both acidic and alkaline media even though a growing body of OER and HER catalysts containing abundant, inexpensive transition metals and

their oxides have recently appeared in the literature [9,10]. The continuous use of relatively expensive materials by industry is because the economics of electrolytic hydrogen production are governed mainly by catalytic performance rather than the catalyst cost and earth-abundant materials are typical not as stable [11]. Recently, Stoerzinger et al. have reported the surface orientated oxygen evolution activities of IrO<sub>2</sub> and RuO<sub>2</sub> in O<sub>2</sub>-saturated 0.1 M KOH electrolyte at 1.53 V versus reversible hydrogen electrode [12]. The potentials range of IrO<sub>2</sub> electrocatalysts for OER were reported from 0.77 V to 1.5 V using acidic and alkaline solutions as the electrolytes [7,10,13,14,15,16,17,18,19]. The lowest potential of IrO<sub>2</sub> nanostructures was reported of 0.77 V for the OER in 0.1 M phosphate buffer solution using ITO electrode [13].

IrO<sub>2</sub> nanostructures have also been used widely as the electrocatalysts for the OER in proton exchange membrane (PEM) water electrolysis [15,16,20], solid polymer electrolyte (SPE) electrolyzer [13,17], photocatalytic and electrolysis of water oxidation, etc [3,6,8,13]. IrO<sub>2</sub> nanostructures were also used as the electrocatalysts in the oxygen reduction [1], oxidation of carbon mono-oxide, methanol and ethanol, etc [21,22]. Apart from the electrocatalysis, IrO<sub>2</sub> nanostructures were also used in several other applications including sensing [23], field emission [24,25], Li-air battery [26], electrical properties, etc [27].

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$\text{IrO}_2$  nanostructured materials were mostly synthesized by metal organic chemical vapor deposition (MOCVD) method [23,28,29,30,31,32,33,34,25]. A variety of  $\text{IrO}_2$  nanostructures have also been produced by various other techniques, including vapor phase transport process [35], electrochemical synthesis [13], arc vaporization [36], hydrothermal [4], reactive radio frequency magnetron sputtering (RFMS) [37], sol-gel [38,39], wetness method [9], Adams fusion method [15,16], sulfite complex route [14], thermal decomposition of precursor ( $\text{H}_2\text{IrCl}_6$ ) [17], oleyl-amine-mediated synthesis, etc [10]. With a closer look, it can be found that many of these methods employ expensive, complex, unstable and not-environmentally-friendly iridium-containing precursors followed with tedious synthetic procedure in complicated experimental setups. Molten salt synthesis process is the cost effective, stable and eco-friendly method compared to above reported methods. This method has also been employed earlier as a simple and facile approach in the synthesis of various nanostructured materials [40,41,42]. Herein, we report the successful synthesis of  $\text{IrO}_2$  nanorods with an average diameter of  $\sim 15$  nm and length of  $\sim 200$  nm for the first time using iridium tetrachloride as precursor in a molten salt of sodium chloride and potassium chloride mixture at  $650^\circ\text{C}$  for 12 h in ambient air atmosphere. The structural and morphological characterizations of  $\text{IrO}_2$  NRs were investigated by powder X-ray diffraction (PXRD), Raman spectroscopy, energy dispersive X-ray studies, single point BET surface area measurements and electron microscopic studies. Note that an effective electrocatalyst can lower the over-potential to sustain the current and, therefore, to improve the efficiency of an electrocatalyst toward the gas evolution reactions due to loss of minimum energy in the electrochemical reactions. Electrocatalytic studies of the  $\text{IrO}_2$  NRs toward OER and HER have been investigated by cyclic voltammetry (CV) and chronoamperometry (CA) in alkaline 0.5 M KOH aqueous electrolyte versus Ag/AgCl electrode, and obtained results were compared to commercial  $\text{IrO}_2$  NPs and previously reported work.

## 2. Experimental

### 2.1. Materials and method

The following reagents were used in the synthesis of the  $\text{IrO}_2$  NRs by the molten salt process:  $\text{IrCl}_4$  (Alfa Aesar, 99.95%), NaCl (Alfa Aesar, 99.0%), and KCl (Sigma Aldrich, 99.0%). The precursors were taken with the molar ratio of 1:30:30 of  $\text{IrCl}_4$ :NaCl:KCl and ground together in an agate mortar pestle for 20 minutes. The resulting mixture was transferred to a covered rectangular crucible and kept at  $650^\circ\text{C}$  for 12 h in a programmable high temperature furnace with the heating and cooling rate of  $5^\circ\text{C}/\text{minute}$  in air. The resulting products were washed with de-ionized water several times and then dried at  $60^\circ\text{C}$  for 6 h in oven. Commercial  $\text{IrO}_2$  nanoparticles (NPs) from Sigma Aldrich (99.9%) were used to compare the electrocatalytic activity with the as-synthesized  $\text{IrO}_2$  NRs.

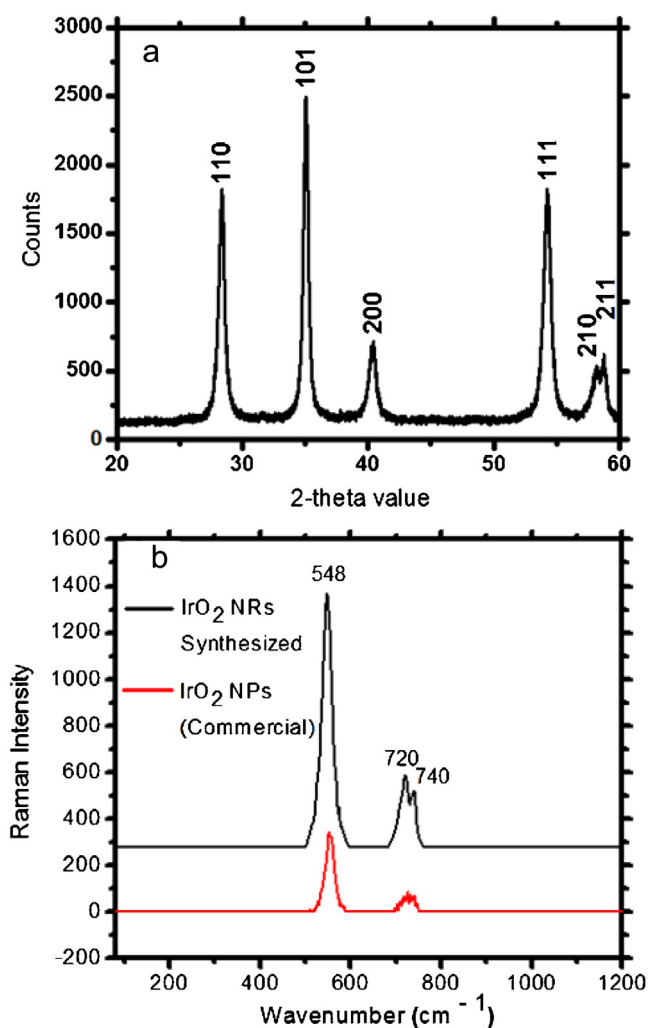
### 2.2. Characterizations

The resulting fine black colored powder of the  $\text{IrO}_2$  NRs was characterized initially by PXRD on a Rigaku MiniFlex X-ray diffractometer using Ni-filtered  $\text{Cu-K}\alpha$  radiation. Diffraction patterns were recorded with a step size and scan speed of  $0.01^\circ$  and 1 s, respectively, on an aluminum sample holder. Raman scattering data was recorded by employing the back-scattering geometries on a Bruker SENTERRA RAMAN microscope with an objective of 20X optical microscope. The excitation line (785 nm) of an Ar<sup>+</sup> laser beam was focused to a spot size of  $5\ \mu\text{m}$  with a laser power of 25 mW. The used spectral resolution range was  $3\text{--}5\ \text{cm}^{-1}$

with the integration time of 100 s. Field emission SEM study was carried out with a ZEISS SIGMA VP electron microscope operated at 5 kV. Transmission electron microscopy (TEM), high resolution TEM (HRTEM), and selected area electron diffraction (SAED) analyses were carried out with a Hitachi H-9500 electron microscope operated at 300 kV. The samples for TEM studies were prepared by dispersing the  $\text{IrO}_2$  NRs in ethanol via ultrasonic process for 5 minutes, and a drop of the formed suspension was put onto a carbon coated copper grid, and then dried in air. Surface area analysis was performed on a Micromeritics ASAP 2020 Surface Area and Porosity Analyzer with nitrogen at cryogenic temperature ( $77.3\ \text{K}$ ).

### 2.3. Preparation of electrode materials and electrochemical measurements

The working electrode was modified with the as-synthesized  $\text{IrO}_2$  NRs for the electrochemical studies toward OER and HER. The  $\text{IrO}_2$  nanostructures modified electrodes were prepared by mixing the nanopowder, carbon black and PVDF. The powder of  $\text{IrO}_2$  NRs was ground with carbon black and PVDF in the weight fraction of 75:15:10, with a small amount of ethanol and then pasted on the glassy carbon electrode. The prepared electrode materials were then dried overnight at  $65^\circ\text{C}$  in vacuum oven. The electrochemical



**Fig. 1.** (a) Powder x-ray diffraction pattern of the  $\text{IrO}_2$  NRs synthesized by the molten salt process at  $650^\circ\text{C}$  for 12 h. (b) Raman spectra of the  $\text{IrO}_2$  NRs and commercial  $\text{IrO}_2$  NPs.

studies were carried out with a potentiostat/galvanostat electrochemical work station (Metrohm Autolab B.V.) at room temperature. The electrochemical work station contains three electrodes including the IrO<sub>2</sub> NRs coated glassy carbon as the working electrode, Ag/AgCl as the reference electrode and a Pt wire as the counter electrode. CV measurements were carried out at different scan rates of 100, 50, 20 and 10 mV/s with a peak window between 0 and -1.4 V versus the Ag/AgCl electrode for the HER and from 0 to 1.0 V versus the Ag/AgCl electrode for the OER. CA characterization was carried out for 200 s at various fixed potentials versus the Ag/AgCl electrode for HER (at -1.4, -1.2, -1.0 and -0.9 V) and OER (at 0.60, 0.65, 0.70 and 0.75 V). CV and CA measurements were carried out in an alkaline electrolyte of 0.5 M KOH aqueous solution. Prior to the electrochemical measurements, nitrogen gas was flown through the electrolyte solution for 5 minutes to remove dissolved air from the electrolyte. Also, electrochemical measurements were run in 0.5 M KOH solution saturated with O<sub>2</sub> toward the OER. Note that freshly prepared electrodes and electrolytes were used in each measurement and the current density of the IrO<sub>2</sub> electrocatalysts was calculated using the geometric electro-active surface area of electrode from Randles-Sevcik equation [43,44,45]. Also note that the geometric electro-active surface area of freshly prepared electrodes could vary due to the variation of amount of pasted materials on fresh glassy carbon electrode. Electrochemical impedance spectroscopic (EIS) studies were also carried in three electrode configuration using the same Metrohm Autolab electrochemical work station with the frequency ranging from 100 mHz to 100 kHz at 1.0 V in 0.5 M KOH solution.

### 3. Results and discussion

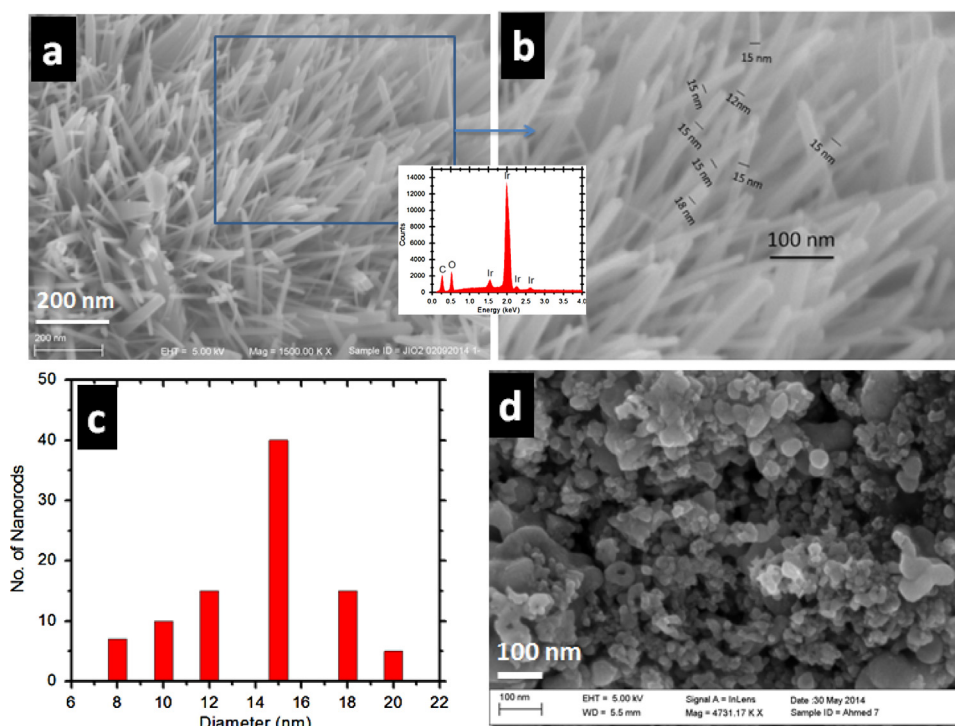
The phase purity of our as-synthesized  $\text{IrO}_2$  NRs was characterized by PXRD and Raman spectroscopy. The observed reflections from the synthesized  $\text{IrO}_2$  NRs (Fig. 1a) could be perfectly indexed with the peaks of (110), (101), (200), (111), (210) and (211) on the basis of rutile phase  $\text{IrO}_2$  with tetragonal crystal

structure (JCPDS file # 88-0288). The relative strong intensity of the (101) peak is an evidence of the orientation of the (101) plane of rutile phase of the  $\text{IrO}_2$  NRs. As detected by PXRD, the commercial  $\text{IrO}_2$  NPs (Sigma Aldrich, 99.9%) were amorphous in nature. In addition to XRD data, evidence regarding the chemical composition of the  $\text{IrO}_2$  nanomaterials was obtained from Raman spectroscopy. Three characteristic Raman bands (Fig. 1b) are identified at 548, 720 and  $740\text{ cm}^{-1}$  from the  $\text{IrO}_2$  NRs and commercial  $\text{IrO}_2$  NPs. Well correlated with those of  $\text{IrO}_2$  nanocrystals, these bands can be correspondingly assigned to the first order  $E_g$ ,  $B_{2g}$ , and  $A_{1g}$  phonon bands of rutile  $\text{IrO}_2$  structure, respectively [37].

Field emission SEM (FESEM) images (Fig. 2a–c) of the as-synthesized IrO<sub>2</sub> product confirm the formation of ultrafine nanorods, appearing in rectangular shape with the diameter ranging from 8 nm to 20 nm and an average diameter of 15 nm. Energy dispersive studies (Inset of Fig. 2a) confirmed that our as-synthesized IrO<sub>2</sub> NRs only contain iridium and oxygen elements with no other impurities, which clearly support the PXRD and Raman data from the IrO<sub>2</sub> NRs (Fig. 1). FESEM image of the commercial IrO<sub>2</sub> NPs shows the spherical shape with an average size of ~60 nm (Fig. 2d).

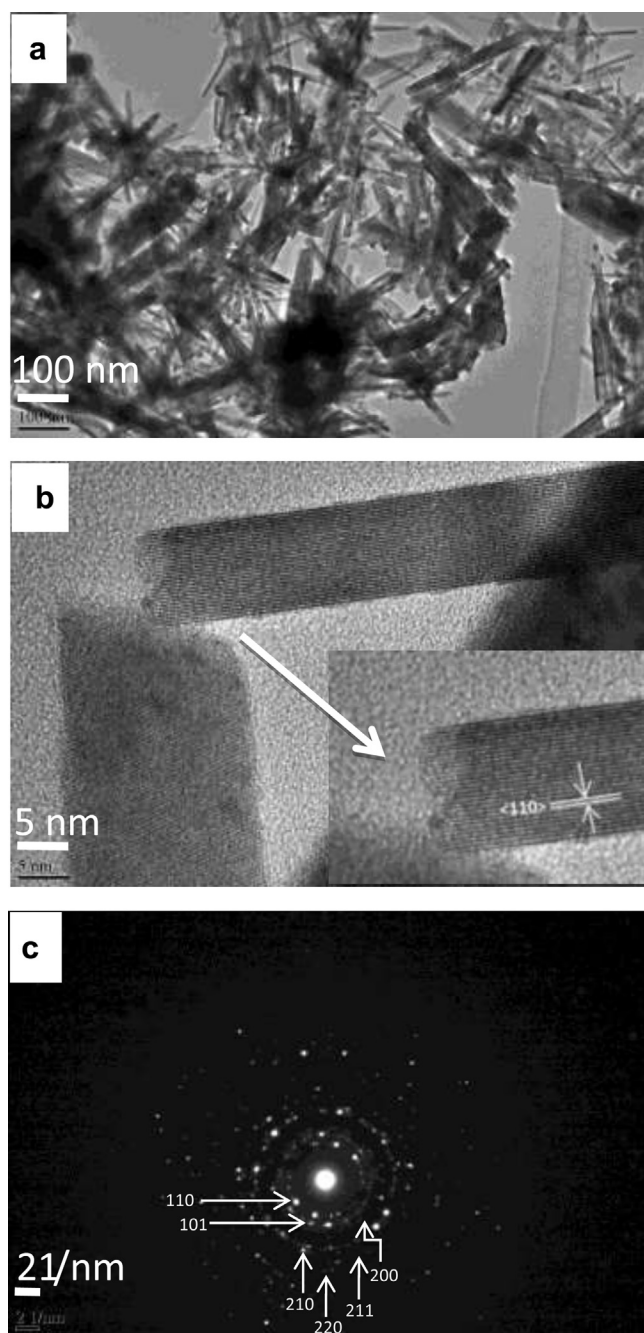
TEM and HRTEM studies (Fig. 3) were employed to characterize the morphology, size and crystalline behavior of the resulting  $\text{IrO}_2$  NRs. TEM image (Fig. 3a) confirms the formation of  $\text{IrO}_2$  NRs with the average diameter and length of  $\sim 15$  nm and  $\sim 200$  nm, respectively. HRTEM image shows a lattice spacing value of  $3.20 \text{ \AA}$ , which is consistent with (110) plane of  $\text{IrO}_2$  NRs (Fig. 3b). Electron diffraction (ED) pattern of the  $\text{IrO}_2$  NRs shows the circular fashion of the spots due to (110), (101), (200), (210), (211), and (220) planes which indicate the formation of tetragonal crystal structure of  $\text{IrO}_2$  NRs (Fig. 3c). Both the HRTEM image and ED pattern confirm the crystalline behavior of the  $\text{IrO}_2$  NRs.

Molten salt synthetic process (i.e. using chlorides as the salts) is a significant approach to synthesize a wide variety of materials with high crystallinity, including the formation of nanospheres,



**Fig. 2.** (a & b) FESEM images of the IrO<sub>2</sub> NRs. The inset shows the corresponding energy dispersive X-ray spectrum. (c) Size distribution of the IrO<sub>2</sub> NRs with respect to the diameters. (d) FESEM image of the commercial IrO<sub>2</sub> NPs.





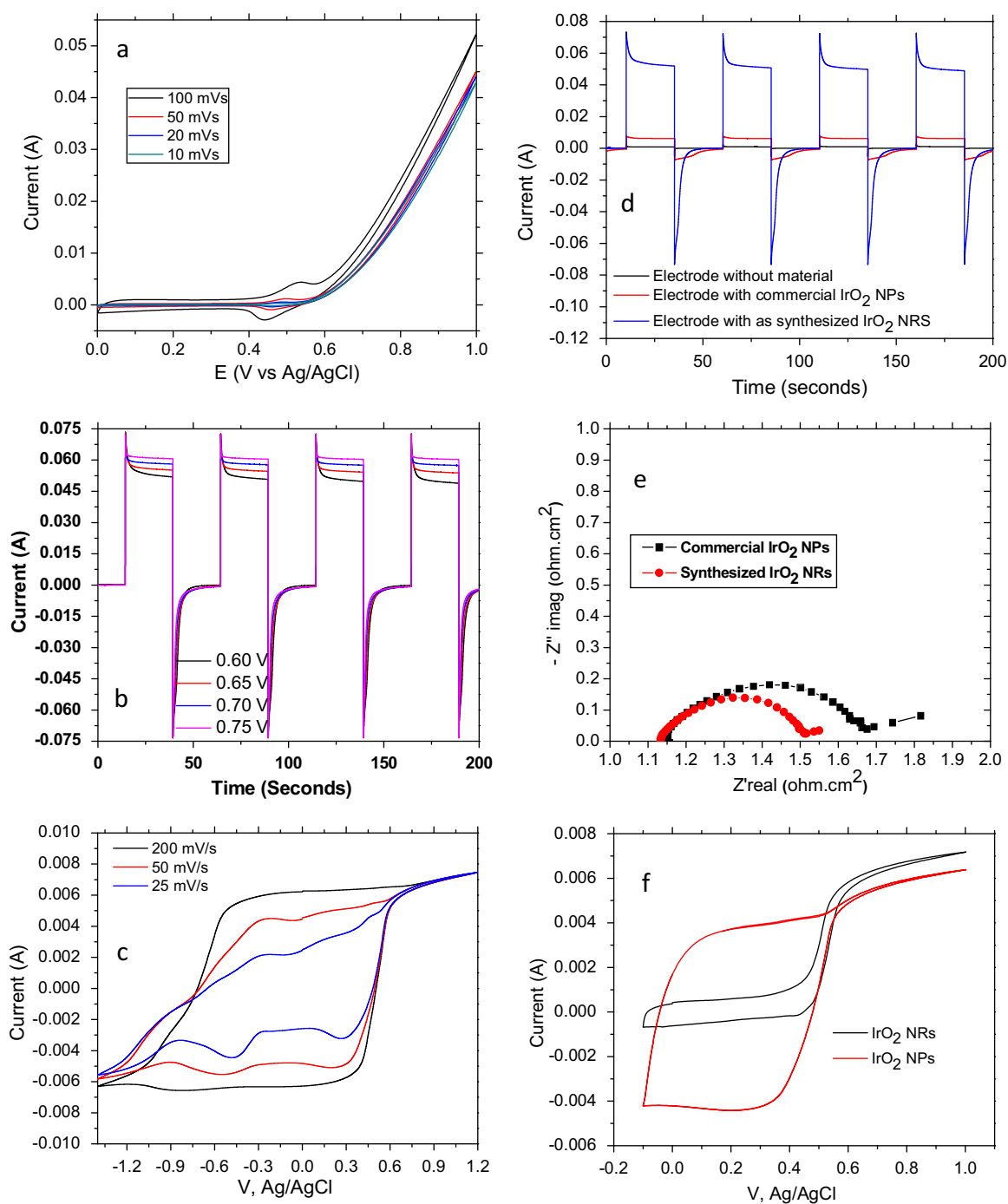
**Fig. 3.** (a) Typical TEM image of the  $\text{IrO}_2$  NRs (scale bar=100 nm). (b) Representative HRTEM image of the  $\text{IrO}_2$  NRs (scale bar=5 nm). (c) Electron diffraction pattern of the  $\text{IrO}_2$  NRs.

nanocubes, and nanorods. The surface and interface energies between the transition metal precursors and the molten salts minimize the activation energies to form unusual morphologies of products [46]. The morphology of the nanomaterials in molten salts could be influenced by various parameters including precursors, synthetic conditions, nature of the salts, diffusion of the salts, and the solubility of the constituents [46,47]. Recently Rørvik et al. had reported that the sodium chloride plays a crucial role in the growth of nanorods of ternary oxides in the molten salt synthesis while no rod shaped morphology was observed without sodium chloride [47]. We observed similar phenomena here from the synthesis of the  $\text{IrO}_2$  NRs with the chloride mixture as the

molten salt. The molten salt method had also been used earlier in the synthesis of nanorods of other binary [48,49], and ternary oxides [47,50,51], but it is the first time for  $\text{IrO}_2$  as reported in this manuscript.

Electrochemical stability and gas evolution activity of the iridium oxides electrocatalysts depend on the chemical and structural nature as well as the size and morphology of the materials and their synthetic conditions [19,52]. Crystalline metal oxides are electrochemically more stable compared to amorphous oxides materials for the gas evolution reactions [19]. However if amorphous  $\text{IrO}_2$  nanoparticles were calcined at high temperature ( $>550^\circ\text{C}$ ) then it could cause the agglomeration of the nanoparticles which may lead to increase their particle size. Therefore, calcined commercial  $\text{IrO}_2$  nanoparticles would not show enhancement of their electrochemical activity towards the OER and HER due to larger particle size. The electrochemical CV and CA measurements were carried out to evaluate the electrocatalytic activity of the  $\text{IrO}_2$  NRs and commercial  $\text{IrO}_2$  NPs toward OER and HER. Fig. 4a shows the cyclic voltammograms of the  $\text{IrO}_2$  NRs with a peak window from 0 to 1.0 V versus Ag/AgCl with scan rates of 100, 50, 20 and 10 mV/s to study their redox behaviors. The peak observed at  $\sim 0.55$  V could be due to the transformation of  $\text{Ir}^{3+}$  to  $\text{Ir}^{4+}$  [10,11]. Noticeably, the electrocatalytic oxidation current was observed beyond this potential in the cyclic voltametric curves due to the following reaction occurs at the electrode surface:  $4\text{OH}^- \rightarrow 2\text{H}_2\text{O} + 2\text{O}_2 + 4\text{e}^-$ .  $\text{IrO}_2$  NPs supported on glassy carbon electrode were also employed for the OER in acidic (0.1 M  $\text{HClO}_4$ ) and alkaline (0.1 M KOH) solutions at 1.5 V versus reversible hydrogen electrode (RHE) by using a rotating ring disc electrode [10]. So far, the lowest potential of  $\text{IrO}_2$  nanostructures for the OER was reported of 0.77 V using the ITO electrode from 0 to 1.4 V versus Ag/AgCl at a scan rate of  $20 \text{ mVs}^{-1}$  in 0.1 M phosphate buffer solution [13]. Herein, this is noteworthy that the OER starts at a much lower potential using our  $\text{IrO}_2$  NRs compared to previous reports [10,13].

Chronoamperometry is a potentiostatic experiment and true quantitative measurements to study the electro-catalytic activity of the materials with three electrode systems at the fixed potential with time. It is important to use less energy for current generation and to find optimal oxygen generating electrocatalysts to minimize the energy loss. Previously, current densities of  $\text{IrO}_2$  nanostructures were reported of  $10 \text{ mA/cm}^2$  at 1.7 V vs  $\text{Hg/Hg}_2\text{SO}_4/\text{K}_2\text{SO}_4$  in 0.1 M KOH and 0.1 M  $\text{HClO}_4$  and  $30 \text{ mA/cm}^2$  at 1.5 V vs  $\text{Hg/Hg}_2\text{SO}_4/\text{K}_2\text{SO}_4$  in 0.1 M  $\text{H}_2\text{SO}_4$  on the literature [10,53].  $\text{IrO}_2$  nanostructures as the electrocatalysts generate a very high current ( $100 \text{ mA/cm}^2$ – $1.7 \text{ A/cm}^2$ ) using PEM and SPE water electrolysis at 1.8 V [14,16,17]. Herein, Fig. 4b shows chronoamperometric measurements of the  $\text{IrO}_2$  NRs supported on glassy carbon electrode at the fixed potentials varied from 0.6 V to 0.75 V. In our experiment, when the potential is switched off then the electrochemical reaction stops and hence current drops drastically. We also observed that the erosion of the materials (from the working electrode) occurred at the higher voltage during OER. Using Randles–Sevcik equation and the geometric electro-active surface area ( $0.70 \text{ cm}^2$ ) of the  $\text{IrO}_2$  NR coated electrode, the current densities were found to be 86, 82, 78 and  $70 \text{ mA/cm}^2$  at the potentials of 0.60, 0.65, 0.70 and 0.75 V, respectively [43,44,45]. Therefore, even our synthesis method to make these  $\text{IrO}_2$  NRs is very facile and cost-efficient using simple precursor, the obtained OER current densities could be comparable with and even higher than what previously have been reported in the literature. Fig. 4c shows the cyclic voltammograms of the commercial  $\text{IrO}_2$  NPs with a peak window from  $-1.4$  to 1.2 V versus Ag/AgCl with scan rates of 200, 50, and 25 mV/s. In CV, an anodic sweep corresponds to the scan towards the positive potential while cathodic sweep denotes the scan towards the negative potential. By comparing the voltammograms from the as-synthesized  $\text{IrO}_2$  NRs with those



**Fig. 4.** (a) CV and (b) chronoamperometric studies of the IrO<sub>2</sub> NRs for OER in 0.5 M KOH deaerated aqueous solution. (c) CV of the commercial IrO<sub>2</sub> NPs in 0.5 M KOH deaerated aqueous solution. (d) Comparative chronoamperometric voltammograms of the as-synthesized IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs for OER at 0.6 V versus Ag/AgCl. (e) EIS of the as-synthesized IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs. (f) CV of the IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs for OER in 0.5 M KOH aqueous solution saturated with O<sub>2</sub> at the scan rate of 50 mV/s.

from commercial IrO<sub>2</sub> NPs for OER, we clearly observe that the IrO<sub>2</sub> NRs show superior electrocatalytic activity over the commercial IrO<sub>2</sub> NPs toward the OER in 0.5 M KOH solution at 0.6 V versus Ag/AgCl electrode (Fig. 4a and c).

Fig. 4d shows the chronoamperometric voltammogram of the three electrodes coated by the IrO<sub>2</sub> NRs, commercial IrO<sub>2</sub> NPs and no materials. The as-synthesized IrO<sub>2</sub> NRs generate higher current density (70 mA/cm<sup>2</sup>, calculated by using the geometric electroactive surface area of electrode of 0.70 cm<sup>2</sup>) than the commercial IrO<sub>2</sub> NPs (58 mA/cm<sup>2</sup>, calculated by using the geometric electroactive surface area of electrode of 0.12 cm<sup>2</sup>) at 0.6 V versus Ag/AgCl

electrode. The chronoamperometry to the current generation depends on many aspects of electrocatalytic-active materials. For example, the dependence of electro-catalytic activity on morphological and orientation of nanostructured materials has been reported [12,54]. Similarly, for IrO<sub>2</sub> nanostructures, the generated current density depends on both the surface area and morphology as a conjunction of both faradaic and non-faradaic processes. In the faradaic process, charges being transferred across the electrode-electrolyte interface are governed by the Faraday's law, and electrons transferred from the IrO<sub>2</sub> atoms to water molecules to generate hydroxide ions and hydrogen molecules as OER and HER,

respectively. For the non-faradaic process, only the formation of the electrical double layer is motivated. From electron microscopic measurements (Figs. 2 and 3), the diameter and length of the as-synthesized IrO<sub>2</sub> NRs were found to be  $\sim 15$  nm and  $\sim 200$  nm, respectively, and the diameter of the commercial IrO<sub>2</sub> NPs was  $\sim 60$  nm. The surface area of the IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs was measured by single point BET and found to be  $\sim 14$  m<sup>2</sup>/g and  $\sim 12$  m<sup>2</sup>/g, respectively. As expected, the surface area increases with decreasing grain size of the IrO<sub>2</sub> nanostructures. However, other than difference on surface area, different crystallinity (rutile crystal phase vs amorphous phase, Fig. 1) and morphology (nanorod vs nanoparticle) may also play important roles on the variation of OER current density between the IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs. Further investigation is needed on these aspects in the future.

Fig. 4e shows EIS spectra of the IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs in 0.5 M KOH solution. The collected Nyquist plots of the Z' real vs negative Z'' imaginary at 1.0 V vs Ag/AgCl show the observed impedance semicircles from both electrode materials. The ohmic resistance of charge transfer ( $R_{ct}$ ) of the IrO<sub>2</sub> NRs was found to be  $\sim 0.35$  ohm·cm<sup>2</sup>, lower than that of the commercial IrO<sub>2</sub> NPs ( $R_{ct} = \sim 0.52$  ohm·cm<sup>2</sup>) and similar with that of the reported Pt/IrO<sub>2</sub> electrocatalyst [2]. Both the Faradic reactions and electrical double-layer on the electrode surface are represented by  $R_{ct}$ . The impedance response, governed by the faradic gas evolutions, supports the enhanced electrocatalytic activity of IrO<sub>2</sub> NRs over the commercial IrO<sub>2</sub> NPs.

The oxygen evolution activity of IrO<sub>2</sub> electrocatalysts has also been carried out in 0.5 M KOH aqueous solution saturated with O<sub>2</sub> gas. Similarly freshly prepared electrodes and electrolyte solutions were used in three electrode system, but O<sub>2</sub> gas flowed for 30 min before starting CV experiments. The cyclic voltammograms of the IrO<sub>2</sub> nanostructures were collected by applying potential ranging from  $-0.10$  to  $1.0$  V versus Ag/AgCl electrode at the scan rate of 50 mV/s (Fig. 4f). From the CV curves, it can be easily observed that the IrO<sub>2</sub> NRs generate higher current compared to the IrO<sub>2</sub> NPs. The current density of freshly prepared electrodes containing IrO<sub>2</sub> NRs (geometric electroactive surface area of electrode =  $0.10$  cm<sup>2</sup>) and IrO<sub>2</sub> NPs (geometric electroactive surface area of electrode =  $0.30$  cm<sup>2</sup>) were found to be  $71$  mA/cm<sup>2</sup> and  $61$  mA/cm<sup>2</sup>, respectively. These results are consistent with those measured in deaerated electrolyte, and that the electrocatalytic activity of the as-synthesized IrO<sub>2</sub> NRs is higher than the commercial and many previously reported IrO<sub>2</sub> electrocatalysts.

Previously, IrO<sub>2</sub> nanostructured materials have been used as an electrocatalyst with SPE in HER [9]. Meanwhile, hydrogen has been proposed as a potential replacement for our rapidly depleting fossil fuel resources [55]. Boodts and Trasatti were reported two decades ago that IrO<sub>2</sub> is a good electrocatalyst for HER and shown the current density in the order of  $10^{-3}$  A/cm<sup>2</sup> [56]. Cheng et al. had reported that carbon supported IrO<sub>2</sub> and RuO<sub>2</sub> generated current density in the range from  $10^{-6}$  to  $10^{-3}$  A/cm<sup>2</sup> with the scan window from  $-0.3$  to  $0.3$  V using  $0.5$  M H<sub>2</sub>SO<sub>4</sub> as an electrolyte [9]. Therefore, HER performance of the IrO<sub>2</sub> NRs was also evaluated by applying a negative potential versus Ag/AgCl electrode. Fig. 5a shows the CV curves of freshly prepared IrO<sub>2</sub> NR electrodes in deaerated  $0.5$  M KOH solution by nitrogen gas for 5 min by scanning with different rates of 100, 50, 20 and 10 mV/s from 0 to  $-1.4$  V versus Ag/AgCl electrode. It was observed that HERs started from  $-0.80$  V on the surface of working electrode containing the IrO<sub>2</sub> NRs versus Ag/AgCl according to the following reaction:  $2\text{H}_2\text{O} + 2\text{e}^- \rightarrow \text{H}_2 + 2\text{OH}^-$ . Chronoamperometric studies of IrO<sub>2</sub> NRs were also carried out for HER over the surface of IrO<sub>2</sub> electrode in  $0.5$  M KOH at fixed potentials with time (Fig. 5b). The current densities from the chronoamperometric measurements using the electroactive surface area of electrode ( $0.35$  cm<sup>2</sup>) for HER were

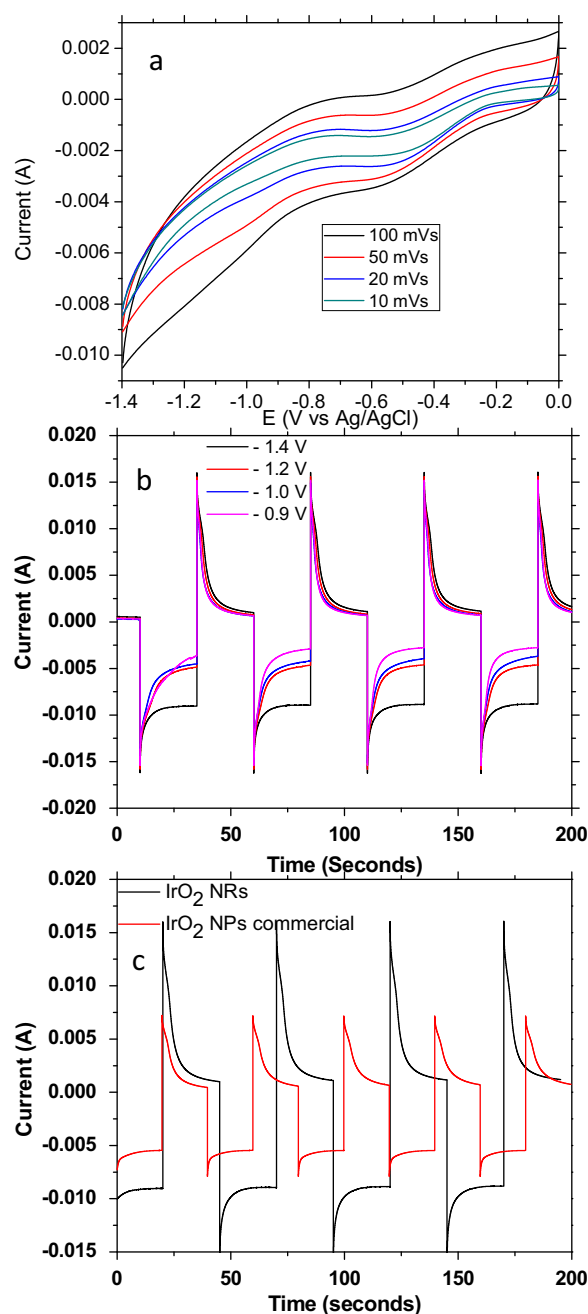


Fig. 5. (a) Cyclic voltammetry and (b) chronoamperometry of the IrO<sub>2</sub> NRs for HER in deaerated  $0.5$  M KOH. (c) Comparative chronoamperometric voltammograms of the as-synthesized IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs for HER at  $-1.4$  V versus Ag/AgCl electrode in  $0.5$  M KOH.

found to be 8, 12, 14 and 25 mA/cm<sup>2</sup> at the potentials of  $-0.90$ ,  $-1.00$ ,  $-1.20$  and  $-1.40$  V, respectively. The generated cathodic current is directly proportional to the rate of hydrogen evolution, so the higher currents generated at  $-1.4$  V may be due to higher electron transfer rate from electrolyte to the electrode surface compared to those at  $-0.90$ ,  $-1.00$ ,  $-1.20$  V. These results are comparable with and even higher than literature results, even direct comparison should be done with caution due to different measurement conditions [9,56]. Fig. 5c shows the comparative chronoamperometric studies of the IrO<sub>2</sub> NRs and commercial IrO<sub>2</sub> NPs at  $-1.4$  V versus Ag/AgCl electrode in  $0.5$  M KOH electrolyte. For HER, the current density of the IrO<sub>2</sub> NRs ( $25$  mA/cm<sup>2</sup>) was



found to be nearly double to that of the commercial IrO<sub>2</sub> NPs (14 mA/cm<sup>2</sup>). Therefore, the IrO<sub>2</sub> NRs show enhanced electrocatalytic HER activity and enhanced OER activity, as discussed above, compared to the commercial IrO<sub>2</sub> NPs and other reported work.

#### 4. Conclusions

In this paper, we report the successful synthesis of ultrafine IrO<sub>2</sub> nanorods with average diameter and length of 15 nm and 200 nm, respectively, using a facile molten salt method at 650 °C in air. These IrO<sub>2</sub> NRs possess enhanced electrocatalytic OER and HER activities compared to the commercial IrO<sub>2</sub> nanoparticles after passing either N<sub>2</sub> or O<sub>2</sub> gas into the electrolyte before experiments, as confirmed by cyclic voltammetry and chronoamperometric measurements and supported by electrochemical impedance spectroscopic studies. Specifically, the as-synthesized IrO<sub>2</sub> NRs generate higher OER current density (70 mA/cm<sup>2</sup>) than the commercial IrO<sub>2</sub> NPs (58 mA/cm<sup>2</sup>) at 0.6 V versus Ag/AgCl electrode in deaerated 0.5 M KOH electrolyte. In 0.5 M KOH electrolyte saturated with O<sub>2</sub>, the OER current density of the IrO<sub>2</sub> NRs and IrO<sub>2</sub> NPs were found to be 71 mA/cm<sup>2</sup> and 61 mA/cm<sup>2</sup>, respectively. Moreover, in 0.5 M KOH deaerated electrolyte, the HER current density of the IrO<sub>2</sub> NRs (25 mA/cm<sup>2</sup>) was found to be nearly double to that of the commercial IrO<sub>2</sub> NPs (14 mA/cm<sup>2</sup>). These current densities are comparable with, and in most cases, higher than reported results in the literature, while the synthesis method employed here for these IrO<sub>2</sub> NRs is much more facile, cost-efficient and environmentally-friendly.

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